

Spallative ablation of dielectrics by X-ray laser

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Abstract Short laser pulse in wide range of wavelengths, from infrared to X-ray, disturbs electron-ion equilibrium and rises pressure in a heated layer. The case where pulse duration τ_L is shorter than acoustic relaxation time t_s is considered in the paper. It is shown that this short pulse may cause thermomechanical phenomena such as spallative ablation *regardless* to wavelength. While the physics of electron-ion relaxation *strongly depends* on wavelength and various electron spectra of substances: there are spectra with an energy gap in semiconductors and dielectrics opposed to gapless continuous spectra in metals. The paper describes entire sequence of thermomechanical processes from expansion, nucleation, foaming, and nanostructuring to spallation with particular attention to spallation by X-ray pulse.

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1 Introduction

There are many industrial applications using short pulse lasers. New exciting possibilities are connected with development of the X-ray lasers. Ablations by the long and short pulses differ qualitatively. The first - evaporate, boil, and, at higher fluences, move matter by ablative pressure created in hot plasma corona. While the release of pressurized layer is the main process in the case of short pulse. It is shown below that this is true for any laser wavelength.

Let's consider short pulse. Irradiation with sufficient intensity transfers substance into warm dense matter state. In condensed phase, the cohesive properties are important. The cohesion is result of interatomic attraction. Due to stiff behavior of solids and liquids, their expansion in rarefaction wave is very different from expansion of gas. Stiff means that the typical moderate

expansions $\Delta\rho/\rho \sim (0.1-0.2)$ cause the order of magnitude pressure drops and even change of sign of pressure - from compressed to stretched state, here $\Delta\rho$ is the density drop due to expansion. Stretched metastable layer under tensile stress appears as a result of the release of the warm dense matter with its stiff response.

In metastable state substances are sensitive to temperature and degree of stretching. Nucleation probability exponentially depends on amplitude of negative pressure. Therefore, sharp nucleation threshold appears on this amplitude. The threshold depends on temperature. Nucleation is followed by development of two-phase layer composed of condensed phase and voids. If the laser heated layer melts, then nucleation takes place in liquid. In this case, expansion of the two-phase layer leads to foaming. The foaming may cause formation of the surface nanorelief [1].

The metastability and nucleation are well-known for release initiated by shock coming to the surface from the bulk of a target. But in the case of shock sent by a long laser pulse, an ion beam, or by an explosion of chemical explosive, near nucleation threshold the two-phase layer locates far from the target surface. In the case of large scale, the development of the two-phase layer cannot disturb the surface of a target because the nucleation layer and the target surface are far from each other and are independent from each other. On the contrary, the short pulse lasers initiate foaming very close to the surface - since the attenuation depth d_{att} for X-ray photons, or thickness of a skin layer δ_{skin} , may be as small as ten nanometers. In this case the foaming strongly interacts with the surface, eventually producing frozen surface structures. Therefore we can use such terms as nanofoam, nanostructures, or nanospallation to describe the situation with small depth d_{att} .

Photon absorption and collisional processes are defined by photon energy and electronic structure. Infrared (IR) and visible radiation excite valent electrons whereas X-rays are absorbed mainly by internal shells in the one-

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photon interactions. This is why the X-ray absorption is qualitatively similar for metals, semiconductors and dielectrics. Action of quanta from the one electron-Volt range of energy depends strongly on existence of the forbidden gap Δ . In metals, they are absorbed mainly via inverse Bremsstrahlung in the skin layer. In cases with gap, the seed Keldysh ionization, inverse Bremsstrahlung heating of ionized electrons and electron avalanche control the rise of number of conduction electrons $n_e(x, t)$ [2, 3, 4, 5]. Significant heating takes place when plasma frequency for these electrons $\omega_{pl}(n_e)$ overcomes laser frequency ω_L during the laser pulse. During the rest of the pulse, the substance with a gap absorbs laser energy in a skin layer. This means that during the rest of the pulse the absorption becomes similar to the absorption of metals. This greatly increases spatial density $J \cdot \text{cm}^{-3}$ of absorbed energy. Estimates show that, in these conditions, laser electric field strength is comparable to atomic fields of external electrons, their wave functions are distorted by electromagnetic wave, and probabilities of the multiphoton and tunnel ionizations are significant.

Relaxation of electrons to equilibrium, after the end of a pulse, depends on a band structure. In metals, free electrons cool due to electron-ion energy transfer and due to thermal conductivity. The same is true for semiconductors which pass to metallic state during their melting. In substances which keep the gap after a pulse, the concentration of free electrons n_e and their temperature T_e decrease after the end of a heating laser pulse as a result of recombination, diffusion of electrons and holes, and electron heat conduction. Three-body recombination is usually more significant than radiative recombination. Relaxation time t_{eq} is of the order of 1-10 ps for all these cases.

An acoustic response time t_s is necessary to decrease pressure few times in the laser-heated layer with thickness d_T . It equals to $t_s = d_T/c_s$ in bulk targets, or $t_s = d_f/c_s$ in foils with small thickness $d_f < d_T$, where c_s is sound velocity.

Electrons are light, their velocities are high and the electronic thermalization time τ_e is small - typically it is at the femtosecond range. In metals, this corresponds to rather high values of $T_e \sim 1 \text{ eV}$. This case is considered here. At room temperatures $T_e \sim 300 \text{ K}$ the e-e relaxation in metals is slow, since e-e collision frequency ν_{ee} is small. If $\tau_e \sim 1 - 10 \text{ fs}$ then at the picosecond time scale we have two thermodynamic subsystems: electrons and ions. Total pressure $p = p_e + p_i$ is composed of partial pressures. The contribution $p_e > 0$ because $T_e - T_i > 0$; in the one-temperature state we have $p_e = 0$.

In our conditions, typical absorbed energy is $E_{abs} \sim (0.1 - 1)E_{coh}$; where E_{coh} is heat of sublimation; e.g., for Al $E_{coh} \approx 3 \text{ eV/atom}$. Then, after e-i relaxation, ion temperatures are in the kiloKelvin (kK) range:

$$T_i \approx 4 (E_{abs}/1\text{eV} \cdot \text{atom}^{-1}) \text{ kK},$$

$$p_i \approx 19 (E_{abs}/1\text{eV} \cdot \text{atom}^{-1})(n/6 \times 10^{22} \text{ cm}^{-3}) \text{ GPa}, \quad (1)$$

if heat capacity is $\approx 3k_B$, and Gruneisen parameter is $\Gamma_i \sim 2$. The parameter $\Gamma = V(\partial p/\partial E)_V$ links pressure rise and fast absorption of energy at the isochoric stage.

At the two-temperature stage $t < t_{eq}$, electron temperatures T_e are much higher than T_i . For metals, in the Fermi-gas approximation, we have $T_e \sim \sqrt{2E_e/\gamma}$,

$$T_e = 17 Z^{-1/6} (n/6 \times 10^{22} \text{ cm}^{-3})^{1/3} (E_e/1\text{eV} \cdot \text{atom}^{-1})^{1/2} \text{ kK}$$

for $t_e = T_e/T_F < 1$, $k_B T_F = E_F$. Here $E_e \sim E_{abs}$ is electron energy per atom, not per electron. Z is the number of electrons per ion, $\gamma = \pi^2 n_e k_B^2 / 2E_F$ is the electron heat capacity constant written in the Fermi-gas approximation.

Electronic contribution p_e is significant when $T_e \gg T_i$, then $T_e - T_i \approx T_e$. In this case, in metals,

$$\frac{p_e}{p_{au}} = \frac{2}{5} Z \frac{n}{n_{au}} \frac{E_F}{E_{au}} \left(\sqrt{1 + t_e^2 \frac{5\pi^2/6 + 25t_e^2/4}{1 + t_e^2}} - 1 \right), \quad (2)$$

where t_e is normalized temperature T_e , and $p_{au} = 2.94 \times 10^4 \text{ GPa}$, $n_{au} = 6.757 \times 10^{24} \text{ cm}^{-3}$, $1/n_{au}^{1/3} = 0.53 \text{ \AA}$ (Bohr radius), $E_{au} = 27.2 \text{ eV}$ are atomic units (au) for pressure, density, etc,

$$E_F/E_{au} = (1/2)(3\pi^2 Z n/n_{au})^{2/3}$$

is Fermi energy. The interpolation (2) has the right limits for $t_e \ll 1$, $t_e \gg 1$.

$$p_e \approx 2.8 Z^{1/3} (n/6 \times 10^{22} \text{ cm}^{-3})^{1/3} (T_e/1\text{eV})^2 \text{ GPa} \quad (3)$$

if $t_e < 1$.

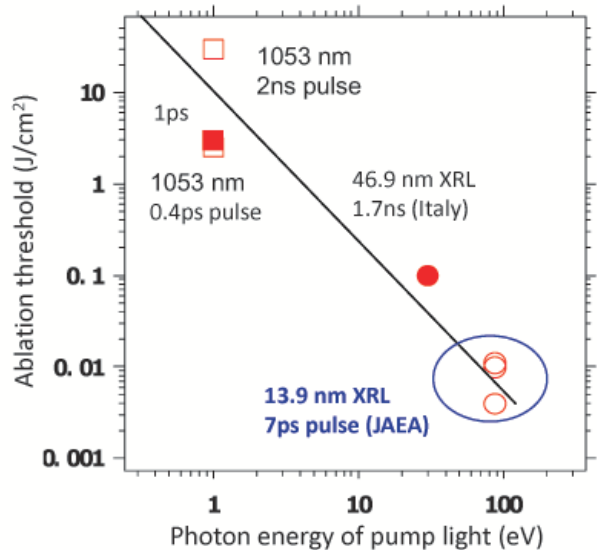


Fig. 1 (Color on line) Comparison of the ablation thresholds (for incident fluence) as function of λ_L and τ_L : three squares [3], filled circle [23], two empty circles - this work.

We call a laser pulse short if $\tau_L < t_s$. Usually the condition $t_{eq} < t_s$ holds. In such a case, the rarefaction release, driven by p_e , envelopes smaller mass than the p_i -release, and nucleation takes place in one-temperature state. In the case when $t_{eq} > t_s$ expansion, subsequent stretching and nucleation are connected with p_e . In the two-temperature state, the vapor-solid and vapor-liquid coexistence curves are shifted in the direction toward the two-phase region. This is a result of a blow out by the electronic pressure of atomic system, composed of atoms which attract each other. At the same absorbed energy, the pressure p_e is few times smaller than p_i , since the degenerate and classic gases of electrons are softer and their Gruneisen parameters $\Gamma_e = 2/3$ is smaller than $\Gamma_i \approx 1.5 - 2.5$ for solids above the Debye temperature.

The main condition for nanostructuring and spallative ablation is $\tau_L < t_s$. Other parameters, such as wavelength (from IR to X-ray) and spectra ($\Delta = 0$ or $\Delta \neq 0$), are less significant. Therefore, spallative ablation is an important mechanism for removal of material. For IR and visible quanta $h\nu \sim 1$ eV acting on metals and semiconductors, theory of spallative ablation is well developed [6,7,8]. The theory is developed to the point where fine and secondary consequences of the metastable decay such as nanostructuring in case of a wide focal spot $d_{spot} \ll d_T$ are predicted [1,9].

For $h\nu \sim 1$ eV and transparent dielectrics, the Newton rings [10,11] were never observed in spite of several attempts [12,13,14]. It seems that interplay of two circumstances makes this observation difficult. Indeed, significant absorption E_{abs} together with small absorption depth, similar to skin depth in metals, are achieved above threshold fluence F_{brkd} for optical breakdown. At the same time, heating E_{abs} is a very sharp function of absorbed fluence F_{abs} near this threshold [2,3,4,5]. Therefore, spallative ablation is limited to the narrow region near F_{brkd} , because the value of heating is restricted $E_{abs} < E_{lim-up}$, $E_{lim-up} \sim 0.3E_{coh}$ [15]. Above this limit, cohesive property is weak against strong stretching in a hydrodynamic rarefaction wave - expansion proceeds similar to expansion of heated gas (that is, without spallative plate). The cohesive property is responsible for creation of a spallative plate and spallative cupola [16]. The cupola is necessary for the interference which results in appearance of the varying in time Newton rings [16].

The second circumstance is connected with the width of the gap Δ . The semiconductors such as Si and GaAs, with rather narrow Δ , metallize during melting. In the meanwhile molten dielectrics remain in dielectric state. In this case, the cupola is dielectric, and Newton interference oscillations are weak (weak oscillations due to presence of oxide film have been detected in [13]).

The Newton rings are bright manifestation of existence of spallative ablation. Appearance of rings means that a light wave interferes between spallative cupola and the rest of the target. This is a very surprising example

of spallative plate so thin that it is even transparent (!) to light - the skin depth is of the order of 10 nm. In more customary spallation by a long laser pulse [17,18], the plate is much thicker.

Contrary to the case with quanta $h\nu \sim 1$ eV and non-transparent substances like metals and semiconductors, the thermomechanical effects for X-ray irradiation are not investigated. This is the first attempt to study thermomechanics, metastability and nucleation induced by X-rays. This subject is interesting in connection with fast progress in developing of X-ray lasers and in connection with a whole number of experimental papers on the X-ray ablation [19,20,21,22,23].

The paper is organized as follows: First, we show how the ratio τ_L/t_s influences the maximum pressure created by absorption of laser energy. Then, new results concerning freezing of nanostructures at a late stage are presented. After that, the theoretical model of spallative X-ray ablation, and experimental findings connected with this model, are described. It is shown that as a result of the conditions, presented in the following lines (i), (ii) and (iii), the threshold F_{abl} for the X-ray spallative ablation is extremely low, in comparison with other cases with different laser wavelength λ_L and durations τ_L . This is illustrated in Fig. 1. The three mentioned conditions are: (i) 100% absorption of X-rays (no reflection); (ii) negligible diffusion and heat conduction loses out from the small attenuation depth d_{att} ; and (iii) smaller energy densities necessary for spallative ablation, in comparison with evaporative ablation of equal amount of material.

2 Duration of a laser pulse and amplitude of an acoustic response

Fast laser pulse transfers matter into energy containing state similar to a state of a chemical explosive behind a front of a detonation wave. This transfer is a base for hydrodynamic release, metastable decay and spallation. As was said, the pulse is short if its duration τ_L is comparable or shorter than acoustic time t_s . In this case, a pathway at a thermodynamic phase plane consists of two parts. One is passed during a pulse τ_L , while the other - during an acoustic response t_s . The first part corresponds to the heating along the isochor $\rho = \rho_{initial}$ with complications concerning the two-temperature details. The second part is formed by an approximately isentropic release along an adiabatic curve. It intersects the coexisting curve and penetrates into the two-phase region.

The states of material near the target surface, irradiated by a long pulse $\tau_L \gg t_s$, are located near the coexistence curve. The hotter state relates to the larger absorbed laser intensity. Pressure created by a long pulse equals approximately to the pressure $p_{sat-vap}$ of saturated vapor. The later is limited by pressure p_{cr} in the

critical point, e.g., for Al $p_{cr} \sim 0.4$ GPa. The critical pressure is significantly below pressures (1), (3) achieved during spallative ablation.

For the same final temperatures $T_{short|bin}$ and $T_{long|bin}$ of material at the coexisting curve, in cases with short and long pulses, the maximum pressure $p_{short|max}$ at the short pulse thermodynamic pathway is much higher; here the subscript "bin" marks the coexistence curve also called the binodal. A short pulse pathway deviates significantly from the coexistence curve into the high pressure condensed phase region. This is why the pressure $p_{short|max}$ is higher. The $T_{short|bin}$ corresponds to intersection of the part 2 of the short pulse pathway with the coexistence curve. The first paper, where the pathway has been used, was the paper [11]. Now this useful conception is widespread [1,6,8,24].

The temperature $T_{short|bin|abl}$ corresponding to spallative ablation threshold is significantly below the critical temperature T_{cr} . Increase of the absorbed fluence F_{abs} above F_{abl} , increases $T_{short|bin}$ above $T_{bin|abl}$. There are distinct fluence F_{ev} and temperature $T_{bin|ev}$ above which the spallative layer disappears [16]. The ratio $T_{cr}/T_{bin|ev}$ depends on material properties and details of the two-temperature stage. It seems that it is larger for Au in comparison with Al.

For $F > F_{ev}$, material expands without spallation plate. This regime is different from the "phase explosion" [25] by long pulse with $T_{long|bin} \approx T_{cr}$. Release of high pressure $p > 0$, created isochorically by short pulse, produces expansion with high rate of stretching $\partial u/\partial x$, where u and x are along expansion direction. After nucleation in metastable state inside two-phase region the inertia of expanding matter inflates bubbles. The inertia is significant since the rate $\partial u/\partial x$ is high. Whereas in the case of the "phase explosion", the expansion of two-phase mixture is driven more slowly by weaker forces connected with pressure difference $\sim p_{sat-vap} - p_{out}$ between pressure inside bubbles and pressure outside the target surface.

Importance of the ratio τ_L/t_s is illustrated in Fig. 2. The acoustic time for this case is $t_s \approx 100$ nm/(5.4 km/s) = 20 ps. We see that positive and negative pressures begin to decrease in their amplitude when the ratio τ_L/t_s becomes larger than unity. Distributions of pressure in the bulk Al target are shown at the instant $t = 10$ ps after the maximum of intensity $I \propto \exp(-t^2/\tau_L^2)$ of pump pulse. We use two-temperature hydrodynamic code described in Ref. [26]. Relaxation time t_{eq} is of the order of 3 ps.

3 Freezing of nanostructures

As previously stated, a short pulse initiates a sequence of stages: (i) release, (ii) metastable state, (iii) nucleation, (iv) evolution of foam. Nucleation takes place if absorbed fluence is above nucleation threshold F_{nucl} . It

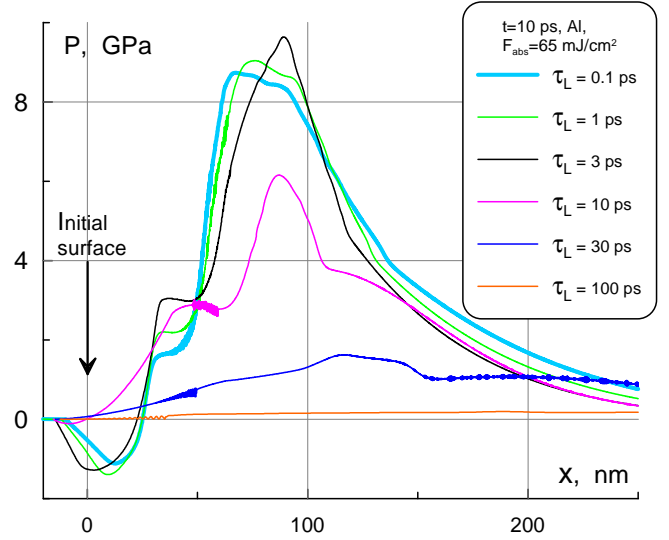


Fig. 2 (Color on line) Strong decrease of maximum pressure for long pulses.

is important that nucleation does not mean tacitly that spallative plate will run away. The spallative ablation threshold F_{abl} and F_{nucl} are separated. The separation $(F_{abl} - F_{nucl})/F_{abl}$ belongs to the few % range. As was shown in [1], during release, the foam inflates to thickness comparable to the distance between the foam and free surface of a target. Therefore it perturbs surface causing appearance of the surface nanorelief.

Development of the nanorelief is a dynamical phenomenon, caused by deceleration of free surface, inflation of foam, and surface tension resistance to inflation. There is the appearance time t_{appr} when it develops. This time is much larger than t_s because fluence F is $\approx F_{abl}$, and the nanorelief appears near the stopping point of free surface. Expansion velocities at this stage are small, and the nanorelief develops slowly. Formation of the nanorelief may be experimentally observed as changes in reflectivity. There are changes due to surface nanorelief and due to absorption in thick foam under surface [27].

In Al, and in many other materials, the ablation threshold F_{abl} is higher than the melting threshold F_{melt} , and then at $F > F_{nucl}$ bubbles appears inside the molten layer. There are two cases with slow development of foam: one when $F_{nucl} < F_{abs} < F_{abl}$ and another when F_{abs} is slightly above F_{abl} . In the first case, the foam remains closed under the surface. An example of this is shown in Fig. 3. In the second case, there is process of slow detachment of the spallative plate. The plate is connected by the random net of the liquid filaments with foam. The filaments are stretched and break off one after another during detachment of the plate. During this process, a nanobrush from a forest of standing filaments appears.

The subsequent evolution is material and target (bulk target versus foil, spot radial size) dependent. Let's com-

ment first the dependence on material properties. In heavy metals late stages of foam development are especially slow. This is significant because there is competition between the late links of the foam development cascade and the rate of conductive cooling. E.g., density and material strength ratios for Au vs. Al are 19.3/2.7 and 20/13 [7], and gold foam moves slower. In Au, the layer between foam and melting front is thinner than in Al, freezing temperature is higher 1337/933, molten layer is thicker, and thermal conductivity 318/237 is higher; while heat capacity and surface temperature 2.5 kK at ablation threshold are approximately equal.

Cooling due to electron heat conduction is important. In liquid state, the foam under target surface $F < F_{abl}$ and nanobrush $F > F_{abl}$ finally disappears because the surface tension collapses bubbles with low vapor pressure in the foam and smooths down the threads and surface bumps. Bubbles collapse when supporting them tensile stress decreases to zero. Contrary to this smoothing, the undersurface bubbles $F < F_{abl}$ and "nano-brush" $F > F_{abl}$ remain in the final relief if conductive losses are fast enough to freeze them. The nano-brush may be accompanied with frozen bubbles under. Comparison of Au vs. Al shows that in the case of Au, this is easier. This picture describes appearance of black gold in recent experiments [1, 28]. Focusing of the laser beam and thickness of foil are also significant, since tight focusing improves cooling, while in thinner foils the reservoir to adopt heat from molten layer is smaller.

Molecular dynamic (MD) simulation of development of foam and its freezing requires huge computer power and smart multi-processors algorithm, because foam occupies large volume and inflation/freezing processes are very slow. In the work [1] the first part concerning the inflation of foam has been done. Here, we present new results describing freezing.

In order to include the electron thermal conductivity in the MD code, the electrons as classical particles were added to atom subsystem. Each "electron" is assumed to associate with its host atom/ion. An electron always allocates the position of host, but it has its own velocity. The neighboring atoms may exchange their electrons with some frequency, determined from known experimental thermal conductivity of liquid aluminum/gold at melting point. The pair of atoms, where electron-electron exchange takes place, is chosen randomly from the list of neighbors of each atom. The list contains neighboring atoms within cutoff distance.

In addition to the frequency of electron-electron exchange between atoms, we adjusted the mass of "electrons", in order to obtain the characteristic time of electron-ion energy exchange in collisions. The electron-ion scattering depends on the velocity vectors of an electron with mass m_1 as a projectile and target ion with mass m_2 according to equations:

$$\mathbf{v}'_1 = (m_1\mathbf{v}_1 + m_2\mathbf{v}_2 + m_2v\mathbf{n})/(m_1 + m_2) \quad (4)$$

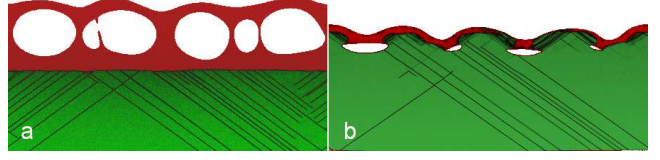


Fig. 3 (Color on line) Freezing of Al foam as result of conductive cooling. Green and red colors correspond to solid and liquid Al. The melting/recrystallization front moves very slow at the time interval to which the left picture $t = 154$ ps belongs. At the right picture $t = 468$ ps the overcooled liquid layer with $T = 570$ K is partly frozen, the underlying bubbles are strongly deformed but survived.

$$\mathbf{v}'_2 = (m_1\mathbf{v}_1 + m_2\mathbf{v}_2 - m_1v\mathbf{n})/(m_1 + m_2) \quad (5)$$

where v is an electron-ion relative speed, and \mathbf{v}'_1 and \mathbf{v}'_2 are the velocities of particles in the laboratory system after collision. The unknown unit vector \mathbf{n} is assumed to be distributed isotropically and is produced by using a generator of random directions. The Eqs. (4,5) conserve the momentum and total energy of electron and ion subsystems. Moreover, such a combined Monte-Carlo-MD approach guarantees conservation of local charge neutrality.

Our tests indicate, that the described above electron-electron exchange and electron-ion collision procedures lead to energy diffusion through dynamic net of atoms, simulated by MD, and give a correct solution of continuum heat conduction equation for liquid aluminum/gold.

Two stages of evolution of Al nanobubble chain are shown in Fig. 3. Fluence in this case is slightly below spallative ablation threshold F_{abl} . Therefore spallative plate keeps its connection to target after the process of inflation of bubbles and freezing. Our approach differs from the approach applied in the important paper [29], where the one-dimensional (1D) finite-difference scheme has been used. The scheme takes into account 1D thermal conductivity and is solved parallel to MD simulation. The 1D scheme employs transversally averaged temperature fields from the parallel MD simulation.

It is important that in case of foam our method allows to describe spatial separation of thermal fluxes pumping heat through threads and walls with small cross-section. This greatly delays cooling through foam. Therefore the base of the foam, which is located from the bulk side, cools faster than the film between the foam and free surface as it is shown in Fig. 3 (b). A liquid layer above the bubbles in Fig. 3 (a) has temperature gradient from 1380 K on its free surface to 1340 K at its internal surface contacting with bubbles. The liquid layer shown in Fig. 3 (b) is in supercooled state, comp. with [30]. Its temperature is $T \approx 570$ K which is below melting point 933.6 K on 360 K, and temperature gradient across the layer is small in this case. Separated nanocrystallites start to grow from two free surfaces of the liquid layer. As 3 indicates a bubble position is not random but correlated with positions of second and third neighbors at

least. The freezing process fixes this correlation with formation of surface nanostructures having correlation length a few inter-bubble distance.

The foam may be partially frozen from the bulk side, when, in the case with fluence above ablation threshold $F_{abs} > F_{abl}$, still liquid film between the foam and free surface of target begins to transform to the spallative plate and the process of breaking of the liquid-solid threads starts. In this case the forest of the frozen threads appears (black gold [1, 28]). MD simulation shows that the gold foam has a thick multi-level structure with significant drop of temperature and expansion rate $\partial u / \partial x$ through levels from one level to another toward the bulk side, see movies [31]. Detailed description of the MD simulation results, concerning foaming and freezing, is beyond the scope of our paper. Here we have to show that a short X-ray laser pulse may cause similar thermomechanical phenomena as an optical pulse. Thick foam is formed if the fluence F is significantly above the threshold F_{abl} . The "flashes" of the successive nucleations during a process of formation of thick foam take place when positive pressure p is lowered by the decrement p_{str} [27], where p_{str} is material strength. At acoustic stage pressure is halved from the initial value to the value corresponding to the established compression wave [7]. If p_{str} does not depend on temperature then the distances between successive flashes are equal. In case $p_{str}(T)$ the distances is growing toward the bulk since temperature decreases and $p_{str}(T)$ increases.

Near threshold F_{abl} the foam evolution time $\sim 0.1 - 1$ ns greatly exceeds collisional relaxation times $\sim t_{eq} \sim 1 - 10$ ps for $h\nu \sim 1$ eV and X-ray lasers. Therefore the picture with melting and foam may be applied to the case of dielectrics heated by short pulse X-ray laser. Similar phenomena may appear in case of metals and semiconductors. In the case of LiF considered below the ablation threshold F_{abl} is lower than the melting threshold F_{melt} . The work [15] is devoted to comparison of the spallative ablations from liquid and solid states. If we rise fluence up to F_{melt} in LiF then deceleration of free surface to almost stopping point will be not possible, because to slow down motion it is necessary to fulfill the condition $F \approx F_{abl}$. This means that in LiF near F_{abl} there are no melting/freezing processes described above. But other substances, e.g. Al, have $F_{melt} < F_{abl}$. In these cases X-ray laser will cause nanostructuring.

4 Theoretical model of X-ray action

Dependence of the attenuation depth $d_{att}(h\nu)$ taken from [32] is shown in Fig. 4. The right arrow corresponds to the Ag X-ray laser described in next Section. The depth d_{att} is large for high energy photons. Then the acoustic response $t_s = d_{att}/c_s$ corresponding to hard photons is long. For energy $\hbar\omega_L = 12$ keV, which will be achieved soon at XFEL [33], it is $t_s \sim 50$ ns. This is

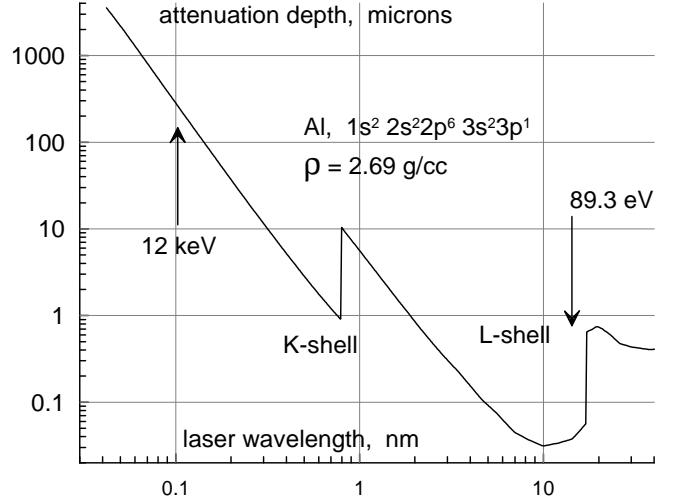


Fig. 4 Variation of $d_{att}(h\nu)$ with photon energy $h\nu$.

very large time in comparison with the picosecond time scale of the atomic processes.

The estimate of the spallative ablation threshold for these hard photons is

$$F_{abl} = \zeta d_{att} n_{at} E_{coh} \approx 200 \text{ J/cm}^2,$$

where $\zeta = 0.2 - 0.4$ is a coefficient [7, 15] in relation $E_{abl} = \zeta E_{coh}$, E_{abl} is energy per atom at the threshold, n_{at} is the atom concentration in solid state. Future XFEL lasers [33] will have $0.1 \text{ keV} < \hbar\omega_L < 12 \text{ keV}$ photons, ultrashort durations $\tau_L \sim 20$ fs, and fluence up to several hundred J/cm^2 . Irradiation by such pulse can cause spallative ablation of huge sub-millimeter piece of condensed target.

Equations describing the electron-ion non-equilibrium stage and hydrodynamic motion are

$$\partial x / \partial t = u, \quad \rho \partial x = \rho^o \partial x^o, \quad \rho^o \partial u / \partial t = -\partial p / \partial x^o, \quad (6)$$

$$\rho^o \frac{\partial E_e^{sum} / \rho}{\partial t} = -\frac{\partial q_e}{\partial x^o} - p_e \frac{\partial u}{\partial x^o} - \frac{\rho^o}{\rho} \dot{E}_{ei} + \frac{\rho^o}{\rho} Q, \quad (7)$$

$$\rho^o \frac{\partial E_i / \rho}{\partial t} = -\frac{\partial q_i}{\partial x^o} - p_i \frac{\partial u}{\partial x^o} + \frac{\rho^o}{\rho} \dot{E}_{ei}, \quad (8)$$

$$\rho^o \frac{\partial n_e / \rho}{\partial t} = -\frac{\partial j}{\partial x^o} + \frac{Q}{u_{i2}} + \nu_{imp} n_e - \kappa_{rec} n_e^3 \quad (9)$$

This is system of equations in Lagrangian coordinate x^o corresponding to an initial position of a material point. The initial density profile inside the target is $\rho(x, t = -\infty) = \rho^o$. Values x , u , $p = p_e + p_i$, $E_e^{sum} = n_e u_{i2} + E_e$, E_i , n_e in (6-9) are functions on variables x^o, t ; n_e is electron concentration n_e in the conduction band; $\dot{E}_{ei} = \alpha(T_e - T_i)$ is an electron-ion energy exchange rate; $q_{e,i} = -(\rho \kappa_{e,i} / \rho^o) \partial T_{e,i} / \partial x^o$ are heat fluxes; $j = -(\rho D / \rho^o) \partial n_e / \partial x^o$ is diffusion flux of electrons; $\kappa_{e,i}$ are electron and ion thermal conductivities.

Primary electrons and primary holes are produced during X-ray pulse acting on LiF. Their kinetic and

potential energies are $E_e|_{prim} \sim 10 - 30$ eV and u_{i1} , $E_e|_{prim} + u_{i1} = h\nu = 89.3$ eV. These energies are measured from the bottom of the conduction band. Primary electrons and holes are non-equilibrium ones - they are described by the distribution function - they can not be described by temperature T_e and concentration n_e . The non-equilibrium electrons and holes relax to thermalized state with secondary electrons and holes through Auger processes, impact ionization, and three-body recombination. Relaxation time is $\tau_{rel} \sim 1$ ps. The value τ_{rel} is smaller than the pulse duration $\tau_L = 7$ ps. Therefore, if we exclude the short time lapse τ_{rel} in the beginning of the laser pulse, then we can include contribution of primary particles into electron energy budget (7) through energy conservation (radiative losses are small). It is supposed in equations (7) and (9) that the laser source $(\rho^o/\rho)Q$ supplies energy directly to secondary electrons and produces secondary electrons at the rate Q/u_{i2} , where $u_{i2} \approx \Delta$ is an ionization potential of secondary electrons, $\Delta \approx 14$ eV is width of forbidden gap in LiF.

Every primary electron produces $(h\nu = 89.3\text{eV})/(u_{i2} + E_e) \approx 6$ secondary electrons. Solution of system (6-9) for parameters $\tau_L = 7$ ps, $F = 10$ mJ/cm², LiF, $d_{att} = 28$ nm gives $n_e|_{max} = (1 - 2\%)n_{at}$, $T_e|_{max} \sim 2$ eV, where $|_{max}$ corresponds to the maximum values. These values are achieved at the end of X-ray laser pulse. The parameters correspond to the experiment described in next Section. Electrons are classical since their kinetic energy $(3/2)k_B T_e \sim 3$ eV is larger than Fermi energy $E_F \sim 0.5$ eV corresponding to our case with low concentration of free electrons.

The energy transfer rate is calculated as $\dot{E}_{ei} = \alpha (T_e - T_i) \approx \alpha T_e = A E_e$, $A = 2\alpha/3k_B$, since heating of a lattice by electrons is significant only when electrons are much hotter than a lattice. Lattice temperature for $F = 10$ mJ/cm² is $T_{at}|_{max} \approx 700$ K. Pressure wave initiated by this fast increase of temperature is shown in Fig. 5. The wave travels to the bulk side of a target (to the right side in Figure) from the irradiated surface, while temperature profile remains "frozen" into matter. The point $x = 0$ is an initial position of the surface. The profile of the wave contains tail with negative pressure p_{neg} (tensile stress). The amplitude $|p_{neg}(t)|$ is enough to cause spallative ablation. Crater depth is 40-50 nm. These findings agree with experimental results described below.

5 X-ray experiment

The experiment has been performed with the Ne-like Ag soft x-ray laser (XRL) facility at JAEA Kansai Photon Science Institute, working at transient collisional scheme [22,34,35]. The XRL beam with an energy ~ 1 microJ and the horizontal and vertical divergences of 12 mrad \times 5 mrad, respectively, was focused on a LiF

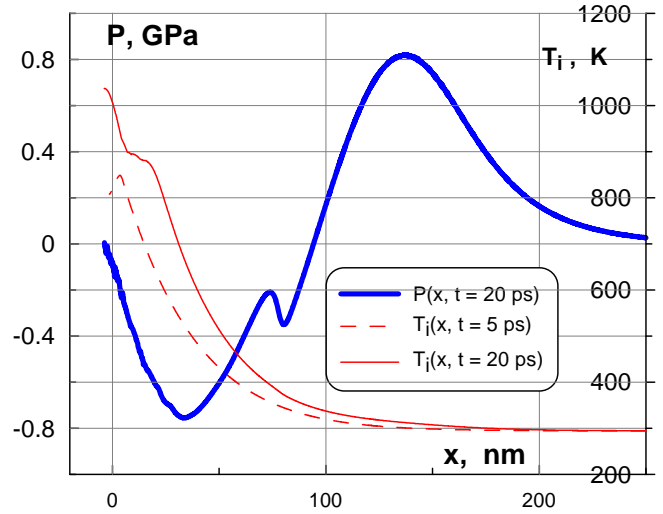


Fig. 5 (Color on line) Pressure and temperature profiles in LiF irradiated by X-ray laser pulse.

crystal of 2 mm thickness and 20 mm diameter, by using a spherical Mo/Si multilayer mirror of 1050 mm radius of curvature (see Fig. 6). The total energy on the LiF crystal of the XRL beam after passing 200 nm Zr filter and reflecting from the focusing mirror was ~ 170 nanoJ in a single shot. The luminescence of stable color centers (CCs) [36,37,38,39] formed by XRL radiation, was used to measure the intensity distribution in the XRL laser focal spot [22,35]. After irradiation of the LiF crystal with the XRL, the photo-luminescence patterns from the color centers (CCs) in LiF (shown in Fig. 6) were observed by using a confocal fluorescence laser microscope (OLYMPUS model FV300). An OLYMPUS BX60 microscope in visible differential mode and an atomic force microscope (AFM, TOPOMETRIX Explorer), operated in the tapping mode, have been used for measurements of the size of ablative spot. As it was shown in our previous experiments [35], only about 6% of full laser energy is concentrated to the best focus spot of ~ 200 square microns. This corresponds to energy ~ 5 mJ/cm² or laser intensity $\sim 7 \times 10^8$ W/cm².

Two types of experimental investigations of XRL ablation threshold of LiF crystals were done. In the first experiments the Zr filter has been removed, and the XRL beam expands without attenuation by the filter. This powerful beam has been focused on the surface of LiF crystal. In Fig. 7 the AFM image of the focal spot of this XRL beam is presented. The image is obtained after a single laser shot. Ablation of crystal is clearly seen at the AFM image and traces. The ablation threshold for LiF irradiated by a single shot is 10.2 mJ/cm². From the traces in Fig. 7, we could see that the ablation depths varied between 30 and 55 nm. These values are close to the theoretical crater depths calculated above.

In the second type of experiments, the Zr filter was settled inside the propagation path of the XRL beam. Three shots have been done in the same focusing spot

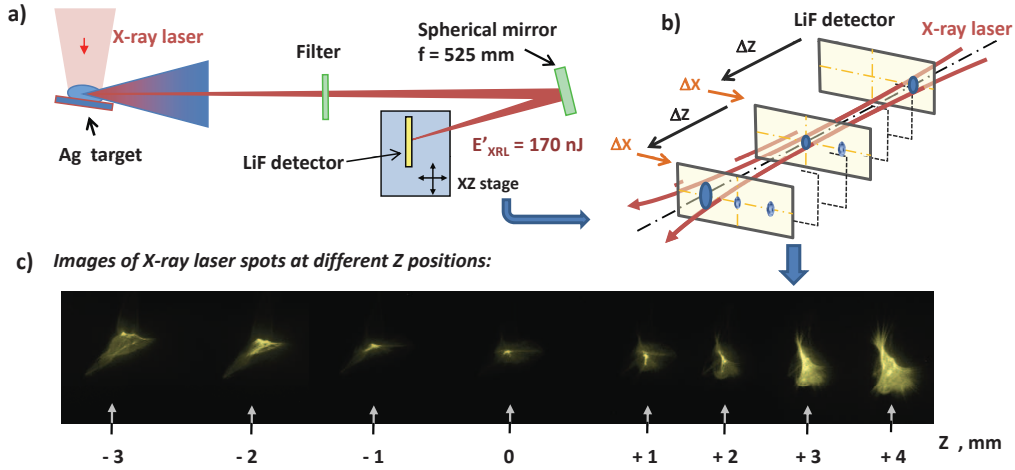


Fig. 6 (Color online) (a) The experimental set up for recording the Ag XRL beam patterns near the best focus position on a LiF crystal. (b) Sketch of motion of a LiF crystal during experiments. (c) The patterns of XRL beam focusing spots recorded on a LiF crystal at -3 mm to +4 mm from the best focus position.

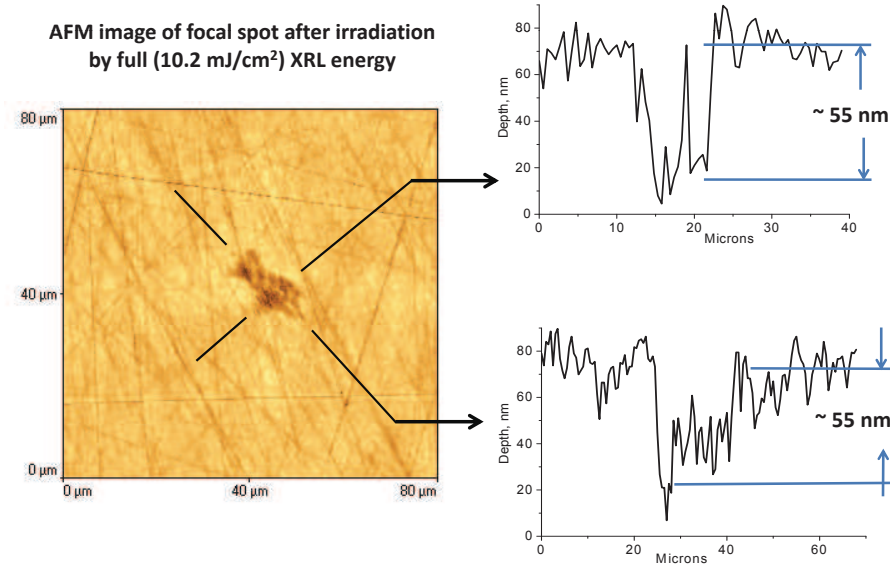


Fig. 7 (Color on line) AFM image and traces, taken through orthogonal directions, of the ablative spot on LiF crystal, irradiated by single shot with full (10.2 mJ/cm^2) laser intensity of XRL beam pulse.

with fluence of 5 mJ/cm^2 for each shot. We could see in Fig. 8 that in this case, a crater, with an ablation depth of about 50 nm, appears on the surface of the crystal. This is similar to the first experiment with single more intensive XRL shot.

The ablation threshold obtained in our experiments is much smaller in comparison with previous experiments, see Fig. 1. Our threshold is 3400, 300, and 10 times smaller than thresholds for nanosecond and femtosecond Ti:sapphire lasers, and for nanosecond 46.9 nm soft XRL, respectively.

6 Conclusion

It is shown that short pulse of XRL causes thermomechanical response as in cases with optical lasers. There is spallative ablation as result of appearance of tensile stress which overcomes material strength above ablation threshold. Near threshold stretching of melt may be accompanied by nanostructuring. Nanostructures freeze down if conductive cooling is fast. Value of the X-ray threshold is small in comparison with irradiation by longer wavelengths and/or longer pulse.

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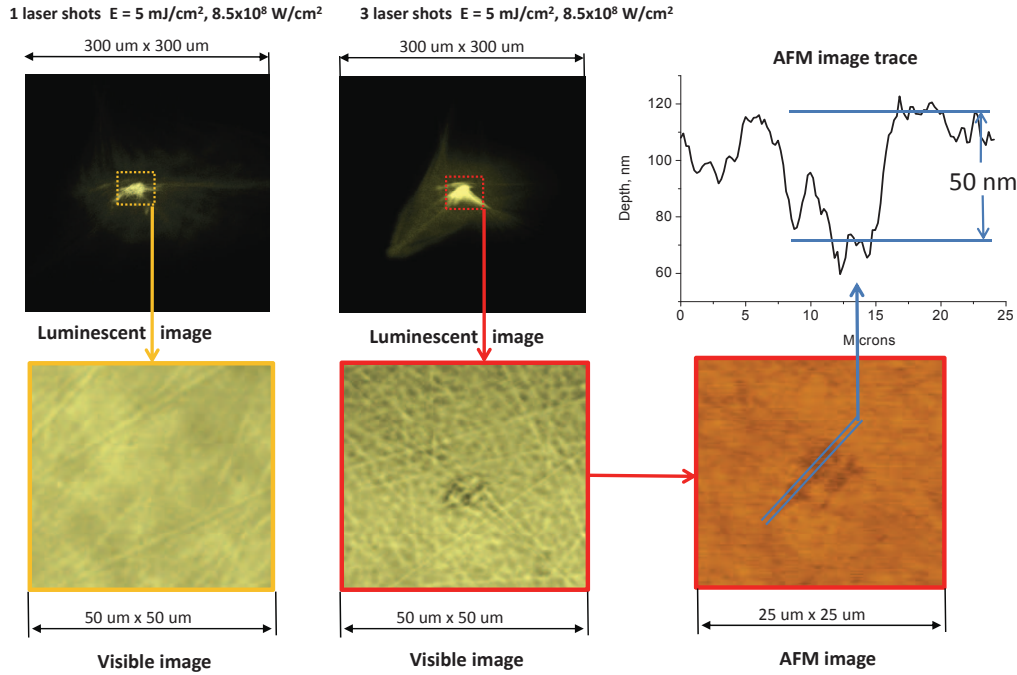


Fig. 8 (Color on line) (a) The luminescence, visible and AFM images of XRL beam focusing spots on the surface of a LiF crystal obtained after one and three shots with XRL laser intensity 5 mJ/cm^2 . Trace was done for AFM image, obtained in the case of three shot irradiation of a LiF crystal.

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